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Final Technical Report

HIGH FIELD CONDUCTION AND SWITCHING PHENOMENA  
AND MAGNETIC SPIN RESONANCE IN METAL OXIDE  
THIN FILMS AND SURFACES

by

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20. ABSTRACT (Continue on reverse side if necessary and identify by block number) Investigation was performed on transport, dielectric, switching and optical properties of thin amorphous and polycrystalline films of a series of transition metal and other oxides films. The conduction and dielectric dispersion mechanisms were unambiguously identified for each oxide studied in the parent metal-oxide-parent metal configuration. The switching phenomena and the observed RF oscillations in the negative resistance region were fully explained by the double injection model. Stable and → next page		

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reproducible switching devices were fabricated and the properties of important switching parameters studied as a function of composition temperature and applied voltage.

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## I PREFACE

This final report briefly reviews the results covering a three year period ending Oct. 31, 1976. The principal objectives of the research program were to study high field conduction properties, dielectric dispersion and switching phenomena in amorphous and polycrystalline metal oxide thin films and their surfaces. Another objective was to improve the switching performance of thin film transition metal oxide devices.

The detailed results of this research program were presented in journal articles, national and international meetings, Ph.D. dissertations and patent applications. As a result of this program 18 papers were published and 4 papers are under preparation or in the submission stage, 8 presentations were made at different meetings, 4 Ph.D theses completed and 2 applications for patentable inventions were made. The list of published papers and other publications is presented in this report.

## II SUMMARY OF RESULTS

### II-1 General Review

An investigation was performed of transport, dielectric and switching properties of amorphous and polycrystalline thin films of NiO, NiO(Li), CoO, CoO(Li), Nb<sub>2</sub>O<sub>5</sub>, TiO<sub>2</sub> and of the optical and electrical properties of SiO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub>, WC<sub>4</sub> and their layered structures with different noble and transition metals.

Experimental results on transport properties of CoO, CoO(Li), Nb<sub>2</sub>O<sub>5</sub> and TiO<sub>2</sub> are found to be in excellent agreement with the generalized theory of the Poole-Frenkel field-assisted emission mechanism. The theory of small polaron hopping mechanism rendered the best agreement with the results of measurements on transport properties of NiO and NiO(Li) polycrystalline films.

Dielectric dispersion was found to be either due to the Debye type of thermally activated hopping mechanism, or to adiabatic hopping from the uncorrelated sites. Relaxation times of different dispersion mechanisms were determined by the use of Cole-Cole or Scaiffe plots. The dielectric breakdown process is fully explained in terms of a formation of four sequential conduction states.

The mechanism of switching from high to low resistance states and of the current controlled negative resistance region was explained on

the basis of I-V characteristics, Balberg's double pulse test and the observed R-f relaxation oscillation in the negative resistance region. The analysis of the above experiments was found to be in excellent agreement with Lampert's double injection theory of switching mechanism. The work on surface switching has shown that devices with a surface electrode configuration resulted in better switching devices. A model and preliminary calculations were made for this type of a device.

Results on the optical, electric and mechanical properties of thin films of  $\text{SiO}_2$ ,  $\text{Al}_2\text{O}_3$  and  $\text{WC}_4$  and their layered structures with noble and transition metals, have shown considerable improvement in mechanical properties and drastic changes in the optical properties of cosputtered metal-oxide films.

## II-2 Transport Mechanism

The I-V-T relationship for a semicrystalline material is given in the generalized Poole-Frenkel field-assisted conduction mechanism by the following expression:

$$I = I_0 (kT)^4 (\beta_{PF})^{-2} \exp(-E_1/kT) \alpha^2 \sinh \alpha$$

where  $\beta = \frac{e^{3/2}}{(\pi \epsilon d)^{1/2}} \frac{1}{kT}$ ,  $E_1$  is the ionization energy and

$$\alpha = \left( \frac{\beta_{PF}}{kT} \right) \left( \frac{V}{d} \right)^{1/2}$$

A plot of  $IT^{-4} \exp(E_1/kT)$  vs  $V^{1/2}/T$  should produce an  $\alpha^2 \sinh \alpha$  curve. This type of a plot was found to be in excellent agreement with the measurements on amorphous and polycrystalline films of  $\text{Nb}_2\text{O}_5$  [1,5,11]  $\text{CoO}$  and  $\text{CoO(Li)}$ [2] and  $\text{TiO}_2$ [10]. The ionization energy  $E_1$  is found to vary between 0.145 eV for  $\text{TiO}_2$  to 0.27 eV for  $\text{Nb}_2\text{O}_5$ .

The I-V and G-V measurements on thin  $\text{NiO}$  and  $\text{NiO(Li)}$  films are found to be in agreement with the following expression for a polaronic hopping conduction mechanism [3,4]:

$$G = \frac{ne^2 a^2 \epsilon'}{kT} v_0 \left[ 1 + \frac{E^2}{2} \left( \frac{ea\epsilon'}{2kT} \right)^2 + \dots \right] \exp \left( - \frac{W_H}{kT} \right)$$

where  $n$  is the free carrier concentration,  $a(4.1\text{\AA})$  is the lattice constant,  $\epsilon' = \frac{E_{\text{local}}}{E} = \frac{\epsilon + 2}{3}$ ,  $\nu \approx 10^{-12}$  sec is the hopping frequency,  $W_H = 0.3$  eV is the activation energy for polaron hopping and  $E$  is the applied field. The experimental plots of  $G$  vs  $V^2$  are found to be in agreement with the above relation.

### II-3 Dielectric Dispersion

Dielectric dispersion and polarization are found to be due to hopping of charge carriers at higher frequencies and to interfacial polarization and blocking electrode effects at low frequencies. The observed frequency dependence of ac conductivity ( $\sigma(\omega)$ )  $\sigma(\omega) \approx \omega$  at higher frequencies [8,10] is in agreement with the theory of non-correlated hopping of charge carriers where:

$$\sigma(\omega) = \left(\frac{\pi^3}{96}\right) N^2(E_F) kT a r_\omega^4 e^{2\omega}$$

Here  $N(E_F)$  is the probability density of localized states at the Fermi level ( $\text{cm}^{-3} \text{eV}^{-1}$ ),  $r_\omega$  is the separation between a pair of localized states ( $r_\omega = a \ln(\nu_0/\omega)$  where  $\nu_0 \approx 10^{12} \text{sec}^{-1}$ ) and  $a$  is the lattice constant. The values of  $N(E_F)$  ranged from  $1.5 \times 10^{19} \text{cm}^{-3} \text{eV}^{-1}$  for  $\text{Nb}_2\text{O}_5$  [8] to  $7 \times 10^{19} \text{cm}^{-3} \text{eV}^{-1}$  for  $\text{TiO}_2$  [10] and  $r_\omega$  from  $49\text{\AA}$  to  $65\text{\AA}$ , respectively.

In the intermediate frequency range, the ac conductivity varies with frequency as  $\sigma(\omega) \propto \omega^n$  where  $0.75 < n < 0.8$  [2,8,16]. This corresponds to the theoretically predicted relationship for non-adiabatic hopping as given by:

$$R_e \sigma(\omega) \propto (14.8 - \frac{1}{2} \ln \omega)^4$$

A plot of the right-hand side of the above relation shows that  $\sigma(\omega) \propto \omega^{0.8}$  which is in close agreement with experimental results.

Relaxation times at low frequencies were obtained from the Cole-Cole plots when modified Debye relations were used in the form [2.8]:

$$\epsilon^* = \epsilon_\alpha + \frac{\epsilon_0 - \epsilon_\alpha}{(1+j\omega\tau)^\beta}$$

with  $\epsilon_1 - \epsilon_\alpha = (\epsilon_0 - \epsilon_\alpha)(\cos\phi)^\beta \cos\beta\phi$

$$\epsilon_2 = (\epsilon_0 - \epsilon_\alpha)(\cos\phi)^\beta \sin\beta\phi$$

Here  $\tan\phi = \omega\tau$  and  $\omega\tau_0 = \tan\theta/\beta$ , where  $\theta = \tan^{-1}\left(\frac{\epsilon_2}{\epsilon_1 - \epsilon_\infty}\right)$ ;  $\tau_0$  is the relaxation time and  $0 < \beta < 1$ . The values for relaxation times ranged from  $2 \times 10^{-2}$  sec [8] to  $1.6 \times 10^{-4}$  sec [2].

#### II-4 Switching Mechanism

Experimental results on switching mechanisms are found to be in excellent agreement with Lampert's theory of double injection mechanism (17). The I-V characteristics show an initial linear increase of I with increasing V followed by an  $I \propto V^2$  relation on increasing the applied voltage. Following the negative resistance region I again becomes proportional to  $V^2$ , which is in complete agreement with Lampert's model.

Experiments on Balberg's double pulse method for testing the switching mechanism rules out the thermal runaway model and was in complete agreement with Lampert's model. The analysis of the observed RF oscillations in the negative resistance region (12) was found to be in agreement with the above model.

#### II-5 Switching Devices

Stable and reproducible switching devices were made of thin transition metal oxide films that could support an input power greater than 2 watts with a device lifetime greater than  $10^8$  switching cycles. The significant and novel features observed in the switching process are the following [7,8,9,11]: a) Switching occurs within  $< 2.5$  mV for triggering pulses in the range 2-4V; b) the delay time  $\tau_d$  decreases exponentially with increasing applied voltage; c) recovery time  $\tau_r$  is 20 n sec and d) the proper switching time  $\tau_s$  is less than 3 n sec.

Switching devices with the surface electrode configuration were studied using NiO and NiO(Li) polycrystalline thin film devices. These devices have shown an increased lifetime by a factor of  $10^3$  switching cycles due to the improved heat exchange with the substrate.

A surface switching electrode configuration, similar to that of a unijunction transistor was studied and it was calculated that the delay time can be reduced below  $10^{-9}$  sec. The experiments, however, were not carried out due to the cessation of the program and the lack of IC facilities.

#### II-6 Optical and Electrical Properties of Other Oxides

These studies were performed on thin films of  $\text{SiO}_2$ ,  $\text{Al}_2\text{O}_3$  and

WC<sub>4</sub>, on the metal-dielectric thin film configuration and on the thin dielectric films codeposited with metals. Noble and transition metals were used in layered structures and codeposition studies. Vastly improved optical and electrical properties were observed in the layered and codeposited films. Since numerous compositions were studied it would be difficult to enumerate them in a report and the details can be found in the references 13, 14 and 15.

### III PERSONNEL

Following is a list of professional personnel who participated in the investigations.

#### Investigators:

- N. Fuschillo, died on Oct. 15, 1974, Professor of Electrical Engineering.
- B. Lalevic, Associate Research Professor of Electrical Engineering.

#### Guest Scientist:

Dr. Bekir Kuliyeve, Azerbaijan Academy of Science.

#### Graduate Students:

N.K. Annamalai, Ph.D.; B. Leung, Ph.D.; W. Slusark, Ph.D.;  
G. Taylor Ph.D.; W. Wang, M.S.

### IV - Ph.D. DISSERTATIONS

1. N. K. Annamalai, "Transport, Dielectric and Switching Properties of Nb<sub>2</sub>O<sub>5</sub> Thin Films in the Device Configuration of Nb - Nb<sub>2</sub>O<sub>5</sub> - Nb" Ph.D. Thesis, Rutgers University, March, 1974.
2. B. Leung, "Transport, Dielectric and Switching Properties of NiO and NiO(Li) Thin Films in the Device Configuration Ni - NiO - Ni", Ph.D. Thesis, Rutgers University, January, 1976.
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4. W. Slusark, "Dielectric Dispersion, Transport and Switching Properties of Thin Polycrystalline Films of Nb<sub>2</sub>O<sub>5</sub>", Ph.D. Thesis, Rutgers University, May, 1976.

### V - PUBLICATIONS

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15. "Hard Transparent Dielectric Coatings," G. Taylor, B. Lalevic and  
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