

INTERIM REPORT

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SUMMARY

Two polyvinyl acetate samples of the Shawinigan Corporation of weight average molecular weights of 85,000 and 190,000 respectively possessing a wide molecular weight distribution but, presumably, a low degree of branching were found to have increased their molecular weights by factors of 2-3 when converted into formals. Osmotic and, in fact, all measurements in ethylene dichloride as solvent present great and as yet unexplained difficulties.

Two polyvinyl acetates of similar molecular weights and of low degree of branching as proven by their hydrolysis and re-acetylation, were supplied by the Bakelite Co. and show an increase in viscosity on butyral formation which, too, has to be interpreted as a sign of molecular weight increase. Here, too, solvent systems more suitable for precise measurements in solution will have to be found, especially as there are indications that the cross-linking process indicated by the molecular weight increase leads to the formation of some microgel particles.

The theory of adsorption of macromolecules with special consideration of their flexibility has been further worked on. Adsorption measurements of formals and butyrals on iron powder of uniform size were started. The first curves obtained show a very steep initial increase in adsorption with polymer concentration followed by a rather flat rise.

FUTURE WORK

The results described above stand in need of considerable broadening. If received in time, two more acetates and butyrals will be characterized. The measurements on all samples are to be extended to more solvents. Efforts to obtain satisfactory osmotic data are to be continued. Some sedimentation work and studies of the dissymmetry of light scattering is under way so that a decision can be made as regards the existence of microgel particles.

The reaction of characterized acetates with phosphoric acid and chromate in organic solvents is under study and this work will be continued. The oxidized products will be re-characterized to establish the changes brought about by the oxidation process. Special attention will be paid to viscosity changes, molecular weight, occurrence of carboxylic or peroxidic groups, and the fate of the acetal ring.

The adsorption studies of oxidized and non-oxidized acetals will be continued and compared with that of other types of high polymers.

August, 1953

POLYTECHNIC INSTITUTE OF BROOKLYN

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INTRODUCTION.

In our first Progress Report we described instrumentation work aiming at the characterization of the polyvinyl acetates (PVAc), polyvinyl alcohols (PVA) and their acetals with a view to elucidating the part played by the polymer structure on the binder efficiency, as explained in our proposal. We reported on some experimental difficulties and on some data on polyvinyl formals (PVF) supplied by the Shawinigan Resins Corporation. Since then we have received polyvinyl acetate samples from this Company which correspond closely to certain of these formals, as well as acetates and butyrals (PVB) from the Bakelite Company. Moreover, we studied more closely the experimental difficulties surrounding our investigation. In fact it appears that they have been underestimated; this is not entirely surprising since no thorough investigation of polyvinyl acetals seems ever to have been undertaken and since an accumulation of three types of groups of various polarity and reactivity on one polymer must tend to complicate solution behavior. In the following we will report on the more important results and observations so far obtained and

on our efforts concerning the implementation of other parts of our program.

EXPERIMENTAL AND RESULTS.

a. Polyvinyl formal.

The measurements of intrinsic viscosities, $[\eta]^*$, and viscosity interaction constants, k' , have been continued (cf. previous report). The data on the Shawinigan polyvinyl acetates were obtained from acetone solutions at 25°C, and are reproduced in Fig. 1a. In addition, weight average molecular weights were obtained from light scattering (Fig. 1b).

The relationship $HC/\mathcal{J} = \frac{1}{M} + 2BC + \dots$ is used to calculate the weight average molecular weight \bar{M}_w . In this equation c is the concentration in gms/ml, \mathcal{J} is the excess scattering of solution above the solvent at 90° angle to the direction of the incident beam, $H = 32\pi^2 n_0^2 / 3N \lambda^4 (dn/dc)^2$ where n_0 is the refractive index of the solvent, N is Avogadro's number, λ is the wavelength of the incident light, and dn/dc the slope of the plot of refractive index versus concentration for the polymer solution. A plot of HC/\mathcal{J} vs. c extrapolated to $c = 0$ is usually linear for uncharged molecules and yields the weight average molecular weight, \bar{M}_w . The dissymmetry of the

* According to an international proposal, the term "Limiting Viscosity Number" is the new name given to the quantity called "Intrinsic Viscosity".

scattering envelope is surprisingly high, indicating either a stiffening of the polymer coil or the presence of some heavily scattering, microgel type, particles. The relevant data are collected in Table I, together with earlier data on the formals in ethylene dichloride.

T A B L E I

Viscosities and Light Scattering Molecular Weights of two Polyvinyl acetates in acetone, and of the derived Formals in ethylene dichloride.

<u>Polymer</u>	<u>Designation</u>	<u>% OAC*</u>	<u>% OH*</u>	<u>[η]</u>	<u>k'</u>	<u>M_w</u>
PVAc	V7	100	---	0.405	0.29	85,000(+10,000)
PVF	7-70	40-45	4.7	0.65	0.33	275,000(+20,000)
PVAc	V15	100	---	0.655	0.37	190,000(+20,000)
PVF	15-95 S	9-13	7-9	0.91	0.62	340,000(+30,000) **

* Data kindly supplied by the Shawinigan Resins Corporation, Springfield, Mass.

** In the previous report the value was misprinted as 34,000

A considerable amount of time has been spent in our attempts to obtain number average molecular weights for the formals from osmotic pressure measurements. However, all efforts to obtain membranes with no or small diffusion of the polymer were so far without success. It does not seem to be possible to condition cellulose membranes in ethylene dichloride

so that the membranes remain sufficiently tight. All osmotic experiments were suffering from diffusion, were poorly reproducible, and yielded molecular weights which were obviously too high.

It was then attempted to tighten the membranes by a technique which had proved successful in other cases. It had been found that high molecular polyvinyl pyrrolidone (PVP) often plugged cellulose membranes on standing. Since PVP is readily soluble in ethylene dichloride, the possibility of a similar action was explored. However, no improvement was found as regards PVF diffusion, and even the osmotic pressures of PVP in ethylene dichloride using pretreated membranes were found to be faulty. It is possible that there is some attack by the solvent on the membrane which increases the pore size, probably by deswelling the texture. Work will be continued by investigating the use of non-cellulose membranes. It should be remembered, however, that ethylene dichloride seems to be a tricky solvent for PVF altogether, since even the light scattering experiments were beset with greater difficulties, than normal, as concerns refractive index determinations, filtration, and general reproducibility. The action of other solvents will be looked into.

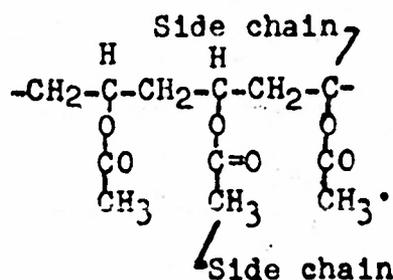
b. Polyvinyl butyral.

We received from the Bakelite Company two samples of polyvinyl butyrals, designated XYSG and XYHL, and two polyvinyl

acetates akin to those used in making these butyrals. The viscosities of the butyrals were measured at 25°C. in methyl ethyl ketone and are shown below; light-scattering molecular weights are being determined.

	<u>$[\eta]$</u>	<u>k'</u>
PVB - XYSG	1.06	.66
PVB - XYHL	.64	.71

Concerning the important question of branching in polyvinyl acetals, the most direct way to estimate branching in the parent polyvinyl acetate is based on its hydrolysis and reacylation of the polyvinyl acetate. This possibility springs from the fact that the branched vinyl acetate polymer has most of the side chains attached to the methyl of the acetyl group, and only a relative few to the carbons on the main chain.



Upon hydrolysis and reacylation, the branches on the acetyl are removed from the polymer molecule. This usually lowers

molecular weight of the polymer but, if the side chains are linked in a ring in the same molecule, hydrolysis and reacetylation will not change the degree of polymerization. Similarly, the measurement of change in molecular weight is not suited for an estimate of the extent of branching which stems from tertiary carbons of the polymer chain. These are not hydrolyzable and not detected by the above procedure. No satisfactory method for the determination of such branches exists, though some information can be obtained from η - M plots for high molecular weights, or from gel point determinations on further polymerization.

Inasmuch as the parent acetates are never exhaustively hydrolyzed, so that some branches may still exist on the remaining acetyl groups, the molecular weights of acetals too may change on thorough hydrolysis, beyond the effect caused by splitting off aldehyde. Apart from this, branching of the acetals will have to be determined, as in the case of other polymers, by comparison of molecular diameters and viscosities of a series of molecular weights.

Since a knowledge of the degree of hydrolyzable branching of the parent acetates remains valuable for the interpretation of the degree of molecular weight change on acetal formation, and of the probability of branches on remaining acetyl groups, the polyvinyl acetate samples XYSG and XYHL were hydrolyzed and reacetylated according to a procedure given

(1)
by Wheeler and co-workers described below.

Two volumes of 5% methanolic solution of the resin were added over a period of 2 hours to 1 volume of a vigorously stirred 1% methanolic KOH solution at room temperature. The agitation was continued for an additional 4 hours. It is essential that the PVA obtained in this step be finely divided. The PVA was then filtered and repeatedly washed with methyl acetate. The drained, but not dry, resin was dispersed in an acetylating solution in the approximate proportions of 40 ml. of solution for each gram of polymer. The acetylating solution comprised 1 volume of pyridine, 5 volumes acetic acid and 15 volumes acetic anhydride. The dispersion was agitated at room temperature for 24 hours longer than required for the resin to dissolve (4-8 days depending upon the state of subdivision of the resin). The solution was dissolved in 2 volumes of benzene and the acetylating solvents removed by repeated extraction with water. The resin was recovered by the boil-out procedure, dried and redissolved in the acetylating solution. After agitating for 2 days at room temperature it was recovered as before, except for 2 additional boil-outs.

The viscosity data on these reacetylated polyvinyl acetates was obtained in two solvent systems, 95% methanol-5% water, and benzene. The results are given in Table II, together with light scattering measurements also carried out in 95% methanol and in benzene.

T A B L E II

Viscosities of original and reacetylated Polyvinyl acetates.

<u>Solution</u>	<u>[η]</u>	<u>k'</u>	<u>M_w</u>	<u>dn/dc</u>
PVAc - XYSG - in 95% methanol	.67	.45	---	.1211
PVAc - XYSG - reacetylated in 95% methanol	.54	.36	---	"
PVAc - XYHL - in 95% methanol	.43	.39	---	"
PVAc - XYHL - reacetylated in 95% methanol	.38	.32	---	"
PVAc - XYSG - in benzene	.935	.19	133,000	.0542
PVAc - XYSG - reacetylated in benzene	---	---	---	"
PVAc - XYHL - in benzene	.588	.24	47,500	"
PVAc - XYHL - reacetylated in benzene	.402	.34	---	"

The data from the reacetylated XYSG, XYHL in 95% methanol show that the plot of $\frac{HC}{\gamma}$ vs. c is not linear at low concentration but tends to curve upward, see Fig. 2. The reasons for this anomaly are not clear; it may be due to residual emulsifying agent, or to polymer aggregation. Whatever the reason, it is certain that here too the solvent system

used is not well suited for molecular weight determinations, and the data were not evaluated.

In benzene the refractive index increment of vinyl acetate is very low (.0542). As the amount of light scattered is proportional to the square of the refractive index increment, the total amount of scattering is low, and the molecular weight determination somewhat inaccurate. The data are given in Table II. Measurements in a more suitable solvent are being conducted.

Adsorption Measurements.

As mentioned briefly in the first report, some measurements of adsorption of polymer from solution were conducted. The polymers used for these experiments were polyvinyl butyral and polyvinyl formal. The adsorbent chosen for these first experiments was an iron powder prepared by Antara Chemicals Division of the General Dyestuff Corporation. This seemed to be a particularly suitable material, because our problem is centered on the adhesion of the acetals to iron and steel. This iron powder is produced by heating iron pentacarbonyl. The particles are almost perfect spheres of extremely pure iron, but with the inevitable oxide surface.

The iron sample used by us is the "SF" type, has a chemical analysis of 98.2-98.8 % iron, 0.50-0.70 % carbon, 0.10-0.13 % oxygen, and 0.55-0.75 % nitrogen. The average

particle size is 3 microns in diameter; less than 5% of the particles are greater than 5 microns and less than 5% of the particles are greater than 1.5 microns in diameter. Some more experiments were conducted using activated carbon and activated alumina.

At the outset of the experiments, it was planned to determine the adsorption of polymer from solution on to solid particles by the measurement of the change in the refractive indices of the polymer solutions. The Rayleigh interferometer which is sensitive to 10^{-6} refractive index units was used for this purpose. In a polymer solvent system with sufficiently high refractive index increment, small changes in concentration could be easily detected.

The polymer solution was shaken up for periods of the order of 12 hours with adsorbents, separated by centrifuging and purified through ultra fine filters; the change in refractive indices was then measured.

About 30 sets of experiments were conducted with the pure solvents alone and some more with solvent-polymer systems. The solvents were pretreated by careful fractional distillation and extraction with the adsorbents, before being used, and the adsorbent was also washed with the solvent.

The measurements obtained were not reproducible, for the pure solvent itself on treatment with the adsorbents showed erratic deviations in refractive indices. It was therefore

decided to try the change in viscosity of a polymer with concentration as the analytical method for detecting the adsorption. After some initial difficulties reproducible results were obtained in the case of polyvinyl butyral XYSG from methyl ethyl ketone on the iron powder mentioned above. Fig. 3 shows the dependence of the amount of adsorption on time for a solution of .265 gm/100 ml concentration. Fig. 4 shows an isotherm obtained with the XYSG sample. The measurements were made at room temperature ($29 \pm 1^\circ\text{C}.$):

In order to obtain a fuller understanding of the process of adsorption of polymers from solution on to solid substrates, it was necessary to further develop the theory of adsorption of polymer molecules. Some work of this character has already been published by us ⁽²⁾ and in addition an investigation proceeding from a different point of view has been ⁽³⁾ undertaken. A report on this is to be presented shortly.

The essential part of this new analysis consists in the construction of a partition function for the adsorbed layer, and partial differentiation with respect to certain of the variables. The calculations are based on the lattice model of polymer solutions.

From this procedure, two equations can be obtained. The first relates the fraction of the surface covered by polymer molecules adsorbed at i points with the fraction of molecules adsorbed at j points. The second equation relates

the fraction of surface covered by polymer molecules at $\frac{1}{2}$ points with the activity of these molecules. The combination of these equations yields the expression for the isotherm. The form of the isotherm has certain similarities with the Langmuir equation, but is considerably more complicated. The course of the curve in Fig. 4 gives an experimental indication of an adsorption curve which is at first steeper and then flatter than a corresponding adsorption curve of low molecular materials. Further comparison of the present theories, with experiment, and extension of the theoretical work so that the model of the adsorbed polymer is more nearly in accord with our ideas of the form of a real polymer chain, constitutes the next steps in this phase of our work.

DISCUSSION.

The molecular weights of the starting Gelvas (Shawinigan PVAc) appear to be somewhat high compared with accepted viscosity conversion figures but the discrepancies are not beyond the range of possible variations. They may be caused, e.g., by a small amount of large and highly branched molecules.

More striking is the increase in molecular weight on converting the PVAc's into the formals. This is somewhat unexpected and indicates that the loss of branching was more than outweighed by aldehyde cross-linkings between molecules.

The larger k' -values, too, are in accord with denser structures. For, if there is on the average at least one half intermolecular bond per molecule, as the weight changes indicate, an appreciably larger number of intramolecular bonds may be expected, which would tend to reduce the volume of the coils. Both inter and intramolecular bonds are certainly in part at the root of the observed solution anomalies; others will spring from the very active solvent ethylene dichloride necessary to dissolve internally netted coils. This possibility will have to be kept in mind when the adhesion properties will be investigated, particularly with a view to the possible differences in the reaction towards the preceding oxidation.

The polyvinyl butyrals also exhibit strikingly high k' -values and the viscosity numbers are also higher than those of the corresponding acetates, although the comparison is not perfect on account of the different solvents. Pending confirmation it appears, therefore, as if the butyrals too had been cross-linked when they were formed from the acetates. The latter were certainly very little branched, as shown by the modest drop in viscosity on hydrolysis and reacetylation, but what amounts in effect to branching has been reintroduced by the intermolecular links formed by the condensing butyr-aldehyde molecules. This branching, as that caused in the formals, is of a rather special type, namely on the average two long branches stemming from a short, schematically shown

FIGURE 1A - VISCOSITY OF POLYVINYL ACETATE IN ACETONE AT 25.0°C.

Samples: V-7
V-15

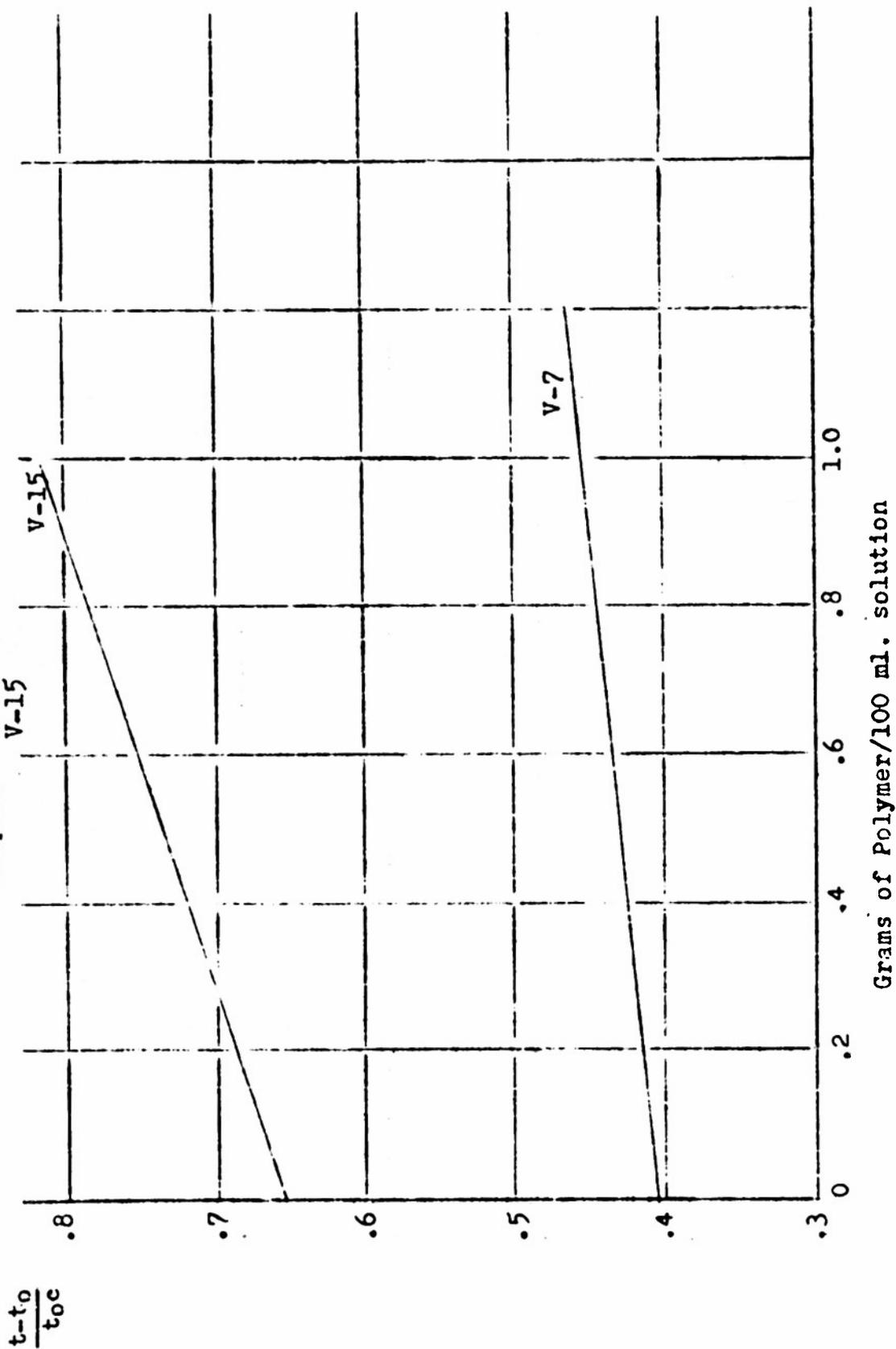
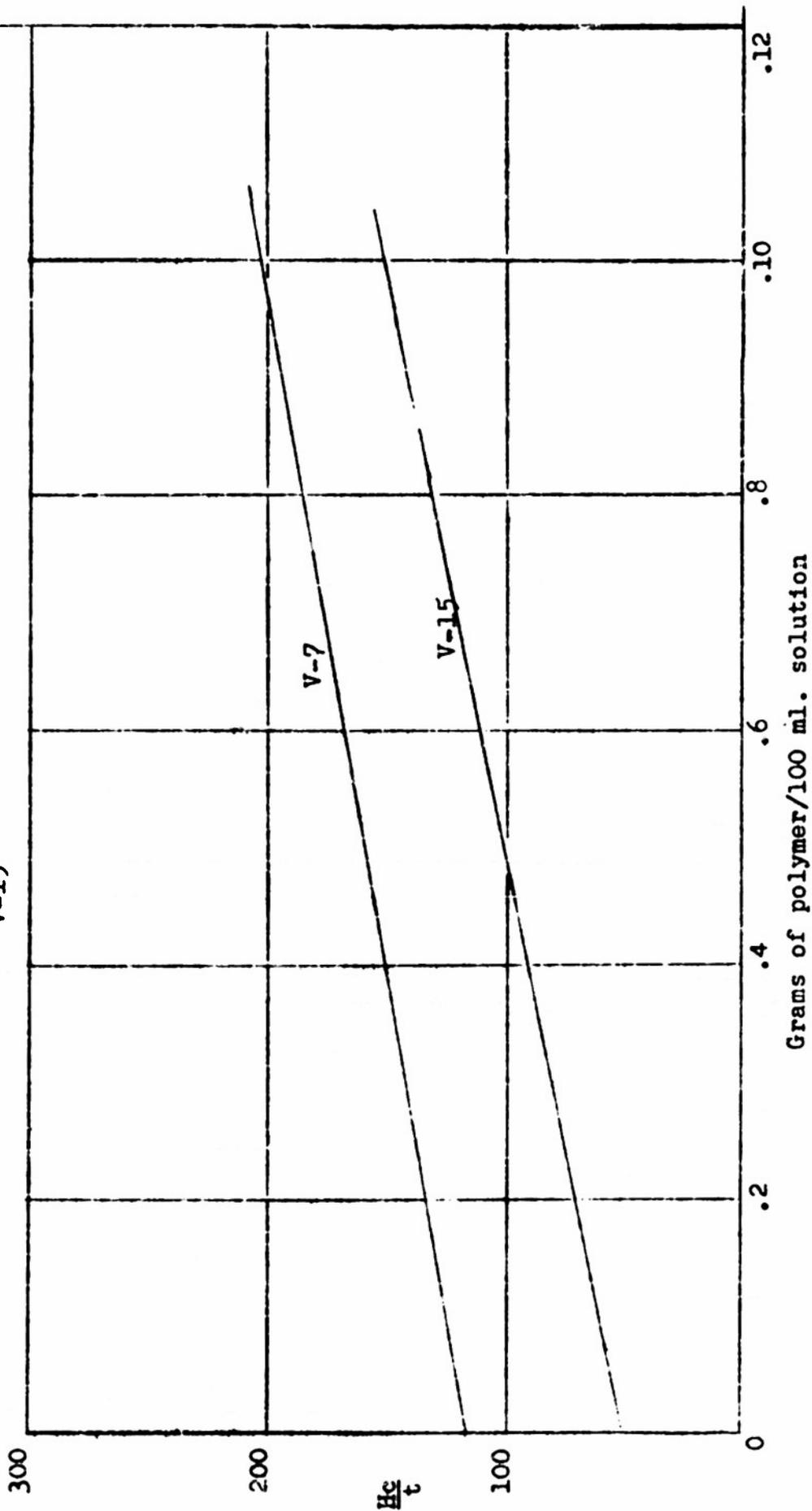


FIGURE 1B. LIGHT SCATTERING OF POLYVINYL ACETATE IN ACETONE

Samples: V-7
V-15



Samples: XYSG
XYSG (reac.)
XYHL

FIGURE 2. LIGHT SCATTERING OF POLYVINYL ACETATE IN 95% METHANOL.

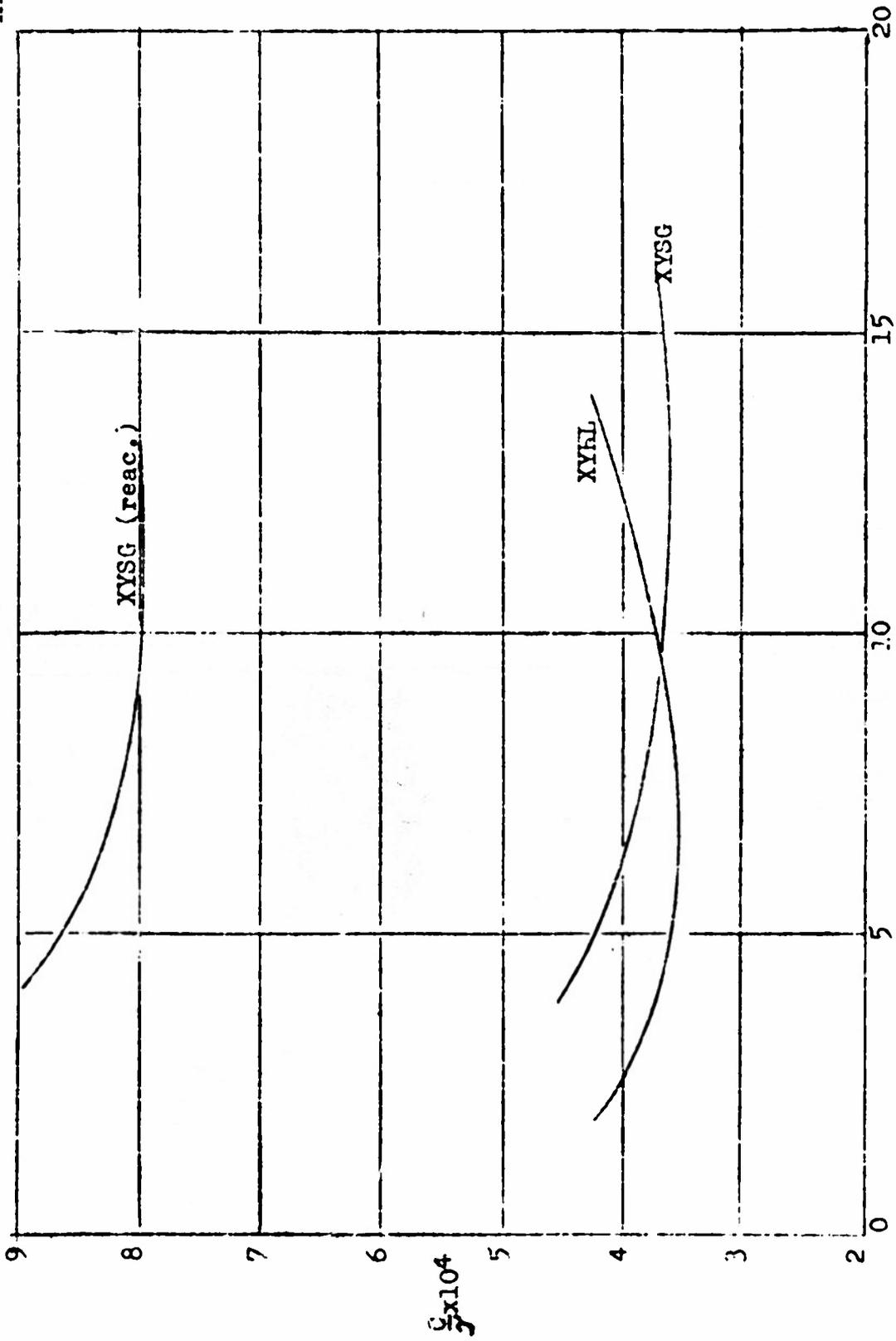
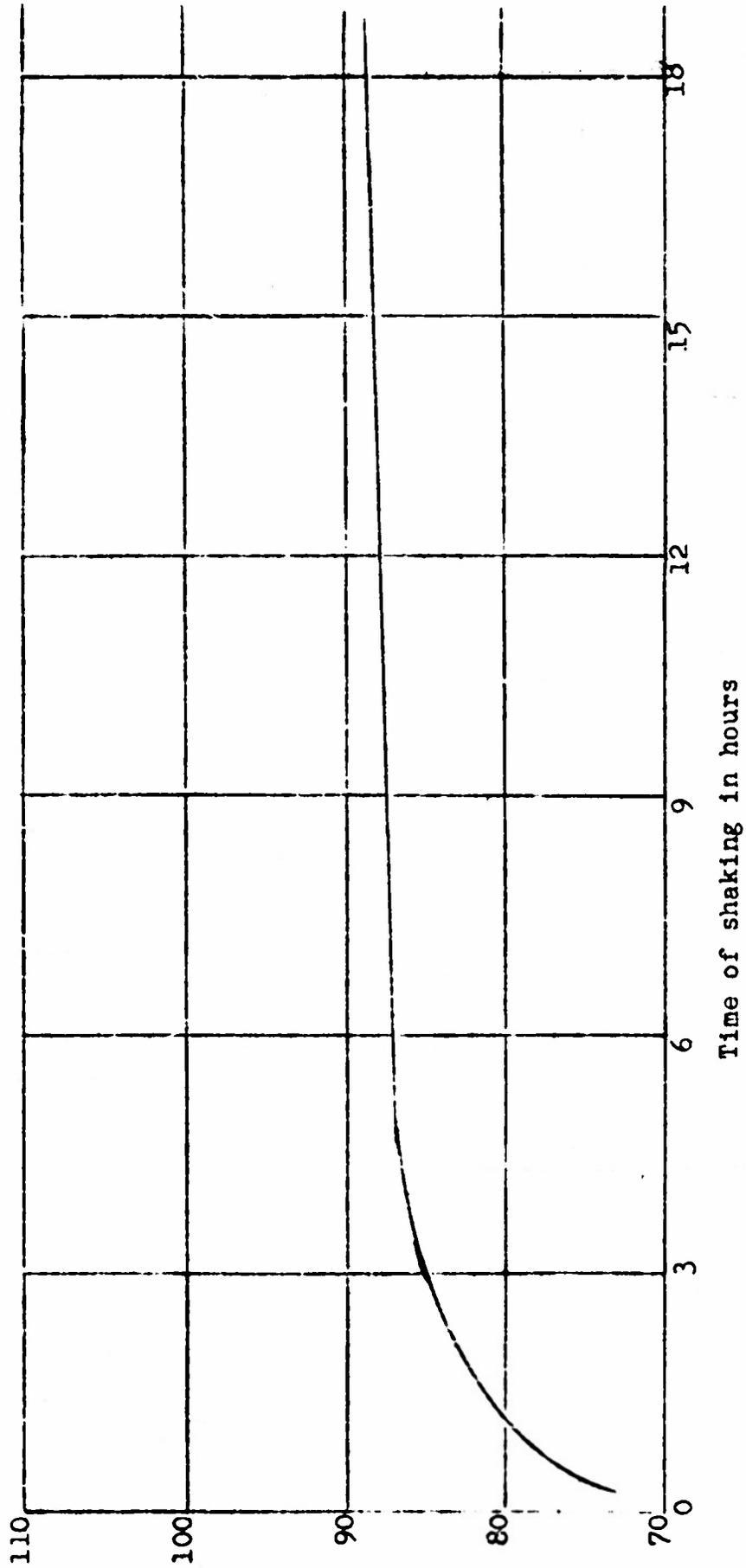


FIGURE 3.

TIME OF SHAKING VS. AMOUNT OF POLYMER ADSORBED.



Flow time of solution - Flow time of solvent in seconds.

FIGURE 4.

ADSORPTION ISOTHERM FOR POLYVINYL BUTYRAL (XYSG)

AT $29 \pm 1^\circ\text{C}$.

